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Mark G. Morgan, Meitian Wang and Arthur Mar*

Department of Chemistry, University of Alberta, Edmonton, AB, Canada T6G 2G2

Correspondence e-mail: arthur.mar@ualberta.ca

## Key indicators

Single-crystal X-ray study
$T=295 \mathrm{~K}$
Mean $\sigma(\mathrm{Si}-\mathrm{O})=0.008 \AA$
$R$ factor $=0.031$
$w R$ factor $=0.074$
Data-to-parameter ratio $=17.9$

For details of how these key indicators were automatically derived from the article, see http://journals.iucr.org/e.
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## Samarium orthosilicate oxyapatite, $\mathrm{Sm}_{5}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}$

Samarium orthosilicate oxyapatite, $\mathrm{Sm}_{5}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}$, contains $\mathrm{SiO}_{4}{ }^{4-}$ tetrahedra $[\mathrm{Si}-\mathrm{O} 1.611$ (7)-1.629 (5) $\AA$ A $]$ and $\mathrm{O}^{2-}$ anions, separated by samarium cations in CN7 [ $\mathrm{Sm}-\mathrm{O}$ 2.2428 (4)-2.641 (7) Å] and CN9 sites [ $\mathrm{Sm}-\mathrm{O} 2.428$ (5)2.846 (5) Å]. The full occupancy of the samarium sites implies a mixed valency of four $\mathrm{Sm}^{3+}$ and one $\mathrm{Sm}^{2+}$ per formula unit.

## Comment

Rare-earth (RE) silicate oxyapatites can be derived from the parent apatite structure, $\mathrm{Ca}_{5}\left(\mathrm{PO}_{4}\right)_{3} \mathrm{~F}$, by substituting $\mathrm{Ca}^{2+}$ with $\mathrm{RE}^{3+}, \mathrm{PO}_{4}{ }^{3-}$ with $\mathrm{SiO}_{4}{ }^{4-}$, and $\mathrm{F}^{-}$with $\mathrm{O}^{2-}$. There are several possibilities for maintaining charge balance. If trivalent RE is assumed, the RE sites must be partially occupied to result in a defect structure, as in $\mathrm{RE}_{4.67}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}(\mathrm{RE}=\mathrm{La}, \mathrm{Sm}$; Kuz'min \& Belov, 1965). Alternatively, if these sites are to remain fully occupied, they must contain a proportion of divalent cations $\mathrm{A}^{2+}$ such as alkaline earths, as in $(\mathrm{RE})_{4}(\mathrm{~A})\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}(\mathrm{Ito}, 1968$; Grisafe \& Hummel, 1970). Yet another possibility is to replace $\mathrm{F}^{-}$, not with $\mathrm{O}^{2-}$, but with $\mathrm{N}^{3-}$, to form the nitridoapatite $\mathrm{Sm}_{5}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{~N}$ (Gaudé et al., 1975). The compound $\mathrm{Sm}_{5}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}$ reported here has a non-defect structure, in contrast to that of $\mathrm{Sm}_{4.67}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}$, previously determined only by photographic methods (Kuz'min \& Belov, 1965).

The structure contains discrete $\mathrm{SiO}_{4}$ tetrahedra (Fig. 1). The Sm 1 atoms are coordinated by O atoms in a pentagonal bipyramidal (CN7) geometry and the Sm 2 atoms in a tricapped trigonal prismatic (CN9) geometry. Atom O4 resides in a characteristic trigonal-planar site, surrounded by


Figure 1
$\mathrm{Sm}_{5}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}$, viewed approximately down the $c$ axis. Colour key: Sm blue, $\mathrm{SiO}_{4}$-tetrahedra yellow, O red.

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Sm 1 atoms. The $\mathrm{Si}-\mathrm{O}$ distances within the tetrahedra (Table 1) are normal $[1.611$ (7)-1.629 (5) $\AA$ ] and agree well with those found in $\mathrm{Sm}_{4.67}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}(1.62-1.63 \AA)$. The fully stoichiometric formula $\mathrm{Sm}_{5}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}$ implies that some divalent Sm must be present, i.e. $\left(\mathrm{Sm}^{3+}\right)_{4}\left(\mathrm{Sm}^{2+}\right)\left(\mathrm{SiO}_{4}{ }^{4-}\right)_{3}\left(\mathrm{O}^{2-}\right)$. This proposal is supported by several observations. First, consistent with the larger ionic radius of $\mathrm{Sm}^{2+}$ relative to $\mathrm{Sm}^{3+}$, the cell parameters in $\mathrm{Sm}_{5}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}[\mathrm{a}=9.4959(10) \AA$ and $c=$ 7.0361 (7) $\AA$ ] are larger than in $\mathrm{Sm}_{4.67}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}(a=9.33 \AA$ and $c=6.85 \AA$ ). The $\mathrm{Sm} 1-\mathrm{O}$ distances $[2.2428(4)-$ $2.641(7) \AA$ a $]$ are similar to those in $\mathrm{Sm}_{4.67}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}$ (2.21$2.67 \AA$ ) but the $\mathrm{Sm} 2-\mathrm{O}$ distances $[2.428$ (5)-2.846 (5) $\AA$ ] are distinctly longer than in $\mathrm{Sm}_{4.67}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}(2.32-2.74 \AA)$. Given the multiplicities of the $\operatorname{Sm} 1(6 h)$ and $\operatorname{Sm} 2$ (4f) sites, an ordered distribution consistent with a $4: 1$ ratio of $\mathrm{Sm}^{3+}: \mathrm{Sm}^{2+}$ is not possible. A reasonable interpretation is that the Sm 1 site contains exclusively $\mathrm{Sm}^{3+}$, and the Sm 2 site contains a mixture of $\mathrm{Sm}^{3+}$ and $\mathrm{Sm}^{2+}$. The bond-valence sums of 2.86 for Sm 1 and 2.54 for Sm 2 are in good agreement with this model (Brese \& O'Keeffe, 1991). Second, an Sr -substituted compound, $\mathrm{Sm}_{4} \mathrm{Sr}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}$, is known (Ito, 1968). Consistent with the similar ionic radii of $\mathrm{Sr}^{2+}$ and $\mathrm{Sm}^{2+}$ (Shannon, 1976), its cell parameters $(a=9.51 \AA$ and $c=7.02 \AA)$ are very close to those of the title compound. Third, the crystals of $\mathrm{Sm}_{5}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O}$ are red, which is characteristic of $\mathrm{Sm}^{2+}$-containing compounds, in contrast to the yellow colour of $\mathrm{Sm}^{3+}$. These crystals were obtained as adventitious byproducts of a reaction in evacuated fused-silica tubes. In contrast to typical preparations of other rare-earth apatites, the low oxygen content under these reaction conditions would be consistent with Sm not being fully oxidized to the trivalent state. A compound having both rare-earth and oxygen defects, $\mathrm{Sm}_{4}\left(\mathrm{SiO}_{4}\right)_{3}$, has been reported; it is red and has cell parameters $[a=9.497$ (3) $\AA$ and $c=$ 6.949 (3) $\AA$ ] that suggest the presence of a small amount of $\mathrm{Sm}^{2+}$ (McCarthy et al., 1967).

## Experimental

A 0.25 g mixture of elemental samarium, gallium, and bismuth in a 1:2:1 ratio (Sm, $99.9 \%$, Cerac; $\mathrm{Ga}, 99.9999 \%$, Alfa-Aesar; Bi , $99.999 \%$, Cerac) was reacted in phase studies of ternary rare-earth bismuth systems. The reactants were heated in an evacuated fusedsilica tube at 1223 K for 4 d , cooled to 773 K over 4 d , and then cooled to room temperature over 1.5 d . Partial attack was observed on the walls of the silica tube, on which were deposited dark-red needleshaped crystals. Windowless semiquantitative EDX (energy-dispersive X-ray) analysis on a Hitachi S-2700 scanning electron microscope revealed the presence of $\mathrm{Sm}, \mathrm{Si}$, and O in these crystals.

## Crystal data

$$
\begin{aligned}
& \mathrm{Sm}_{5}\left(\mathrm{SiO}_{4}\right)_{3} \mathrm{O} \\
& M_{r}=1044.02 \\
& \mathrm{Hexagonal}, P 6_{3} / m \\
& a=9.4959(10) \AA \\
& c=7.0361(7) \AA \\
& V=549.46(10) \AA^{3} \\
& Z=2 \\
& D_{x}=6.310 \mathrm{Mg} \mathrm{~m}^{-3}
\end{aligned}
$$

## Data collection

Bruker Platform/SMART
1000 CCD diffractometer $\omega$ scans ( $0.2^{\circ}$ frames)
Absorption correction: numerical
(SHELXTL; Sheldrick, 1997)
$T_{\text {min }}=0.179, T_{\text {max }}=0.643$
6710 measured reflections

## Refinement

Refinement on $F^{2}$

$$
\begin{aligned}
& w=1 /\left[\sigma^{2}\left(F_{o}{ }^{2}\right)+(0.0365 P)^{2}\right. \\
& +5.6399 P] \\
& \text { where } P=\left(F_{o}{ }^{2}+2 F_{c}{ }^{2}\right) / 3 \\
& (\Delta / \sigma)_{\max }<0.001 \\
& \Delta \rho_{\text {max }}=2.75 \mathrm{e}_{\AA^{-3}} \\
& \Delta \rho_{\min }=-2.73 \mathrm{e}^{-3} \\
& \text { Extinction correction: SHELXL97 } \\
& \text { Extinction coefficient: } 0.0015 \text { (3) }
\end{aligned}
$$

Table 1
Selected interatomic distances ( $\AA$ ).

| $\mathrm{Sm} 1-\mathrm{O} 4$ | $2.2428(4)$ | $\mathrm{Sm} 2-\mathrm{O} 2$ | $2.508(5)$ |
| :--- | :--- | :--- | :--- |
| $\mathrm{Sm} 1-\mathrm{O}^{\mathrm{i}}$ | $2.383(5)$ | $\mathrm{Sm} 2-\mathrm{O} 1$ | $2.846(5)$ |
| $\mathrm{Sm} 1-\mathrm{O}^{\text {ii }}$ | $2.421(6)$ | $\mathrm{Si}-\mathrm{O}^{\mathrm{iv}}$ | $1.611(7)$ |
| $\mathrm{Sm} 1-\mathrm{O} 1$ | $2.497(5)$ | $\mathrm{Si}-\mathrm{O} 2$ | $1.623(7)$ |
| $\mathrm{Sm} 1-\mathrm{O} 3$ | $2.641(7)$ | $\mathrm{Si}-\mathrm{O} 1$ | $1.629(5)$ |
| $\mathrm{Sm} 2-\mathrm{O}^{\text {iii }}$ | $2.428(5)$ |  |  |

Symmetry codes: (i) $y,-x+y, \frac{1}{2}+z$; (ii) $1-y, 1+x-y, z$; (iii) $x-y, x,-z$; (iv) $-y, x-y, z$.

Refinements on the absorption-corrected data were performed, in which the occupancies of successive sites were allowed to vary. These converged to values of 1.00 (4) for $\mathrm{Sm} 1,0.97$ (4) for $\mathrm{Sm} 2,1.00$ (4) for $\mathrm{Si}, 1.00$ (5) for $\mathrm{O} 1,1.12$ (5) for $\mathrm{O} 2,1.07$ (5) for O 3 , and 1.10 (7) for O4, with reasonable displacement parameters for each site. These results support a fully stoichiometric model, and the occupancies were fixed at 1.00 for all atoms in the final refinement. The atomic positions were standardized with the program STRUCTURE TIDY (Gelato \& Parthé, 1987).

Data collection: SMART (Bruker, 1997); cell refinement: SAINT (Bruker, 2000); data reduction: SAINT; program(s) used to solve structure: SHELXTL (Sheldrick, 1997); program(s) used to refine structure: SHELXTL; molecular graphics: ATOMS (Dowty, 1999); software used to prepare material for publication: SHELXTL.

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